Electronic structure of the cubic perovskite $SrMn_{1-x}Fe_xO_3$ investigated by x-ray spectroscopies

J.-S. Kang,^{1[,*](#page-4-0)} H. J. Lee,¹ G. Kim,¹ D. H. Kim,¹ B. Dabrowski,² S. Kolesnik,² Hangil Lee,³ J.-Y. Kim,³ and B. I. Min⁴

1 *Department of Physics, The Catholic University of Korea (CUK), Bucheon 420-743, Republic of Korea*

2 *Department of Physics, Northern Illinois University, DeKalb, Illinois 600115, USA*

³*Pohang Accelerator Laboratory (PAL), POSTECH, Pohang 790-784, Republic of Korea*

⁴*Department of Physics, POSTECH, Pohang 790-784, Republic of Korea*

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The electronic structures of SrMn_{1−*x*}Fe_{*x*}O₃ (0≤*x*≤1) have been investigated by using photoemission spectroscopy (PES), soft x-ray absorption spectroscopy, and soft x-ray magnetic circular dichroism (XMCD). Both Mn and Fe ions are found to be nearly tetravalent for the whole range of *x* in SrMn_{1−x}Fe_xO₃. Valence-band PES measurements show the broader Fe 3*d* $(t_{2g}^3 e_g^1)$ bands than the Mn 3*d* (t_{2g}^3) bands and the finite metallic intensity near E_F in the Fe 3*d* PES of SrFeO₃ in agreement with the metallic ground state of SrFeO₃. The Fe 2*p* XMCD spectra exhibit the enhanced XMCD effect for $0.6 \le x \le 0.8$ in SrMn₁_{*x*}Fe_{*x*}O₃, which is consistent with the spin-glass behavior in the intermediate substitution, while the Mn $2p$ states exhibit the negligible XMCD effect.

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Perovskite $SrFeO₃$ shows the metallic behavior when fully oxygenated.^{1[,2](#page-4-2)} SrFeO₃ exhibits neither the Jahn-Teller distortion^{3[,4](#page-4-4)} nor the orbital ordering,⁵ which is unusual considering that high-spin (HS) Fe^{4+} $(t_{2g}^3 e_g^1)$ ions are usually Jahn-Teller active. Oxygen deficiency in $\text{SrFeO}_{3-\delta}$ affects the transport properties dramatically, causing the metal-toinsulator transition (MIT) for $\delta \ge 0.15.^{4,6-9}$ $\delta \ge 0.15.^{4,6-9}$ $\delta \ge 0.15.^{4,6-9}$ It is likely that the MIT in SrFeO_{3− δ} is accompanied by the charge ordering of mixed-valent Fe^{3+}/Fe^{4+} ions. In this aspect, it is interesting to study SrMn_{1−*x*}Fe_{*x*}O₃ perovskite oxides, which are expected to exhibit competition between various valence states of Mn $(3+$ and 4+) and Fe $(3+, 4+,$ and 5+). Since the $Fe^{4+}(3d^4)$ ion is isoelectronic with the Mn³⁺(3*d*⁴) ion, SrMn_{1−*x*}Fe_{*x*}O₃ with Mn⁴⁺(t_{2g} ³) and Fe⁴⁺(t_{2g} ³ e_g ¹) ions is expected to demonstrate the competing spin, charge, and lattice interactions similar to doped perovskite manganite oxides of $R_{1-x}A_xMnO_3$ [*R*=rare-earth ions and *A*=divalent ions (Ca, Sr, and Ba)], which in particular could lead to ferromagnetism and colossal magnetoresistance.

Both $SrMnO₃$ and $SrFeO₃$ show the antiferromagnetic (AF) ordering with Néel temperatures T_N =233 K for SrMnO₃ (Ref. [10](#page-4-8)) (G type) and T_N =134 K for SrFeO₃ (Ref. [11](#page-4-9)) (spiral type). On a substitution of the cubic perovskite $SrMnO₃$ preserving the valence state 4+ of Mn (e.g., Sr_{1−y}Ca_yMnO₃), the AF ordering is observed in the cubic, tetragonal, and orthorhombic crystal structures.¹² T_N is gradually suppressed by the deviation of the Mn-O-Mn bond angle from 180° and by the variance of the average size of the *A*-site ion via changes in the Sr/Ca ratio.^{12,[13](#page-4-11)} Cubic $SrMn_{1-r}Fe_xO₃$ with 180° bond angles shows rapid suppression of both the *G*- and spiral-type AF orders and exhibits the antiferromagnetic order and/or the spin-glass behavior at an intermediate substitution.⁵ No ferromagnetism or magnetoresistance has been observed.

With the Mn substitution in $SrFeO₃$, the system becomes an insulator. Mössbauer measurements^{5,[14](#page-4-12)} for SrMn_{1−*x*}Fe_{*x*}O₃ revealed that there are two types of Fe ions for $x \le 1$, which were considered to be in the charge disproportionation (CD) of $Fe^{3+}(d^5) + Fe^{5+}(d^3)$. Note that Mössbauer spectroscopy re-

vealed no CD feature for metallic $SrFeO₃$.^{[5,](#page-4-5)[15](#page-4-13)} For isoelectronic charge-ordered (CO) CaFeO₃, however, such a CD feature was observed in Mössbauer spectroscopy[.15](#page-4-13) On the other hand, the analysis of x-ray photoemission spectra¹⁶⁻¹⁸ suggested that the CO state in $CaFeO₃$ results from the CD of $2d^5L \rightarrow d^5L^2 + d^5$ rather than $2d^4 \rightarrow d^3 + d^5$ *(L_i*; an oxygen hole or a ligand hole). Hence, a further study is needed to settle down the issue of Fe valence states in SrMn_{1−*x*}Fe_{*x*}O₃.

In order to better understand the peculiar electronic and magnetic properties of $SrMn_{1-x}Fe_xO_3$, it is essential to investigate the electronic structures of Fe and Mn 3*d* electrons. In this work, we have investigated the valence and spin states of Mn and Fe ions in stoichiometric SrMn_{1−*x*}Fe_{*x*}O₃ (0≤*x*≤1) by using photoemission spectroscopy (PES),^{[19](#page-4-16)} soft x-ray absorption spectroscopy (XAS) , 20,21 20,21 20,21 20,21 and soft x-ray magnetic circular dichroism $(XMCD).^{22,23}$ $(XMCD).^{22,23}$ $(XMCD).^{22,23}$ $(XMCD).^{22,23}$ These methods provide valuable information on the valence states and the electronic structures of solids.

Polycrystalline samples of SrMn_{1-*x*}Fe_{*x*}O₃ (0≤*x*≤1) were prepared by using a two-step synthesis methods as described in Ref. [5.](#page-4-5) X-ray powder-diffraction (XRD) and neutron powder-diffraction measurements revealed no impurity phase. PES, XAS, and XMCD experiments were performed at the 2A undulator beamlines of Pohang Accelerator Laboratory (PAL) at two different experimental chambers. The 2A beamline is an elliptically polarized undulator, from which the circularly polarized light was obtained with the degree of circular polarization of 90*%*. Samples were cleaned *in situ* by repeated scraping with a diamond file under the base pressure of \sim 7×10^{−11} Torr. The data presented in this paper were obtained from the samples nearly free of surface contamination. PES spectra were obtained at room temperature by using a SCIENTA SES100 analyzer. The Fermi level E_F and the overall energy resolution of the system [full width at half maximum (FWHM)] were determined from the Fermi-edge spectrum of scraped Au metal. The FWHM of the valenceband PES spectra was set at \sim 150 meV at $h\nu \sim 600$ eV. XAS spectra were obtained both at room temperature and at $T \sim 80$ K (Ref. [24](#page-4-21)) by employing the total electron yield

FIG. 1. (Color online) Comparison of the Mn 2p XAS spectra of $SrMn_{1-x}Fe_xO₃$ with those of MnO₂ (Mn⁴⁺), Mn₂O₃ (Mn³⁺), and MnO (Mn^{2+}) .

(TEY) mode with the photon energy resolution of \sim 100 meV at $h\nu$ \sim 700 eV.

XMCD measurements were done at $T \sim 80$ K under the applied magnetic field of ~ 0.7 T. XMCD spectra were taken for a fixed helicity of light by reversing the applied magnetic field at each $h\nu$. In order to minimize the artificial effects caused by the decreasing photon flux with time, the direction of the applied magnetic field was reversed at each data point in the XMCD data acquisition. XMCD spectra were obtained by employing the TEY mode with the photon energy resolution of \sim 120 meV at $h\nu \sim$ 700 eV. All the PES, XAS, and XMCD spectra were normalized to the incident photon flux.

Figure $1(a)$ $1(a)$ shows the measured Mn 2 p XAS spectra of $SrMn_{1-x}Fe_xO_3$ for $x=0$, 0.2, 0.4, 0.6, and 0.8. As the amount of substituted Fe increases, the energy positions of the *L*³ $(2p_{3/2})$ and L_2 $(2p_{1/2})$ peaks do not change and their line shapes remain similar. These findings indicate that the valence state of Mn does not change with *x*. As *x* increases, the FWHM of the Mn *L*³ peak decreases—resulting in the narrowing of the L_3 peak and, concomitantly, the more pronounced low-energy shoulder with increasing *x* (hv \sim 641 eV). This trend is shown more clearly at the top of Fig. [1](#page-1-0)(b), where two XAS spectra of $x=0.8$ (SrMn_{0.2}Fe_{0.8}O₃) and $x=0$ (SrMnO₃) are scaled to have the same area. Also note that the intensities of the L_2 peaks of $x=0.8$ and 0 are almost the same in this scale. The more pronounced shoulder structure of the Mn L_3 peak with increasing x implies the weaker covalency of the Mn 3*d* orbitals in the dilute Mn concentration. The weaker covalency the increased localized character) in the dilute Mn concentration could result in the reduction in the charge-transfer effect^{21[,25](#page-4-22)[,26](#page-4-23)} between the Mn 3*d* orbitals and the O 2*p* ligand orbitals (L) from $x=0$ to 0.8. This trend seems to be contrary to the more metallic nature of $SrMn_{1-x}Fe_xO₃$ for increasing *x*, and so the origin of the reduced charge-transfer effect for large *x* is not clear at the moment.

Figure $1(b)$ $1(b)$ compares the Mn 2*p* XAS spectra of SrMnO₃ $(x=0)$ and $SrMn_{0.2}Fe_{0.8}O₃$ *(x=0.8)* with those of reference oxides such as tetravalent MnO_2 $(Mn^{4+} : 3d^3),^{27,28}$ $(Mn^{4+} : 3d^3),^{27,28}$ $(Mn^{4+} : 3d^3),^{27,28}$ $(Mn^{4+} : 3d^3),^{27,28}$ trivalent

FIG. 2. (Color online) Comparison of the Fe 2p XAS spectra of $SrMn_{1-x}Fe_xO_3$ with those of Mn_2O_3 $(Mn^{3+}:3d^4)$ and Fe_2O_3 $(Fe^{3+}:3d^5)$. Here Mn_2O_3 (3*d*⁴) is chosen to represent the isoelectronic $\text{Fe}^{4+}(3d^4)$ ion.

 Mn_2O_3 (Mn^{3+}), 28 28 28 and divalent MnO (Mn^{2+} : $3d^5$). ^{[27](#page-4-24)} The peak positions and the line shapes of both $x=0$ and 0.8 are similar to those of MnO₂ but quite different from those of Mn₂O₃ and MnO, indicating that Mn ions are formally tetravalent $(Mn^{4+}:3d^3)$ in the whole range of substitutions. Some differences between the Mn 2*p* XAS spectra of SrMn_{1−*x*}Fe_{*x*}O₃ and that of $MnO₂$ might be due to the different ligand fields and the different charge-transfer effects between $SrMnO₃$ and MnO₂. The nature of tetravalent Mn⁴⁺ states for all x in SrMn_{1−*x*}Fe_{*x*}O₃ is also supported by the negligible Mn 2*p* XMCD effect in SrMn_{1−*x*}Fe_{*x*}O₃ for all *x* values as shown in Fig. [5.](#page-3-0)

Figure $2(a)$ $2(a)$ shows the measured Fe $2p$ XAS spectra of SrMn_{1−*x*}Fe_{*x*}O₃ for *x*=0.[2](#page-1-1), 0.4, 0.6, 0.8, and 1 and Fig. 2(b) compares the Fe $2p$ XAS spectra of SrFeO₃ with those of reference oxides for trivalent $Fe₂O₃$ (Fe³⁺:3*d*⁵),^{[29](#page-4-26)} divalent FeO $(Fe^{2+}:3d^6)$ $(Fe^{2+}:3d^6)$ $(Fe^{2+}:3d^6)$, ^{[29](#page-4-26)} and Fe metal.³⁰ In Fig. 2(b), we have chosen Mn_2O_3 (Ref. [28](#page-4-25)) to represent the tetravalent (Fe⁴⁺) state because both of them are isoelectronic in the ground state with the $3d^4$ configuration (the L_3 and L_2 parts of the Mn₂O₃ spectrum have been separated and shifted to be aligned to the corresponding L_3 and L_2 peaks of SrFeO₃). The line shape of our Fe $2p$ XAS spectrum for $x=1$ (SrFeO₃) is similar to that in literature. $25,26,31$ $25,26,31$ $25,26,31$

It is clearly shown that the measured Fe 2*p* XAS spectrum of SrFeO₃ is very similar to that of Mn_2O_3 (3d⁴) but different from those of α -Fe₂O₃ and Fe metal, providing evidence that Fe ions are formally tetravalent (Fe^{4+}) in SrFeO₃. This conclusion agrees with the interpretation of the Mössbauer spectrum for $SrFeO₃$ single crystal.⁷ In contrast, two major discrepancies are observed clearly between S rFeO₃ and α -Fe₂O₃ with trivalent Fe ions. First, the low-energy peak (at \sim 707.6 eV) in α -Fe₂O₃ is missing in SrFeO₃. Second, the line shape of the L_2 peak for SrFeO₃ is different from that of α -Fe₂O₃. These discrepancies confirm that Fe ions are not trivalent in SrFeO₃. Note further that the line shape of the Fe $2p$ XAS of SrFeO₃ is very different from that of MnO_2 $(Mn^{4+}:3d^3)$ [see Fig. [1](#page-1-0)(b)]. MnO_2 could be considered to represent the quintvalent Fe⁵⁺ state since Fe⁵⁺ and

FIG. 3. (Color online) (a) Comparison of the valence-band PES spectra of SrMn_{1−*x*}Fe_{*x*}O₃ for *x*=0 and *x*=1. The Mn 3*d* and Fe 3*d* PES spectra represent the partial spectral weight distribution of the Mn and Fe 3*d* electrons, respectively. See the text for the details. (b) Comparison of the O 1*s* XAS spectra of SrMn_{1−*x*}Fe_{*x*}O₃. Each feature is identified in the figure. PES and O 1*s* XAS spectra were obtained at room temperature.

 Mn^{4+} ions have the isoelectronic $3d^3$ configuration. These differences provide unambiguous evidence for the absence of either Fe³⁺ or Fe⁵⁺ ions in SrFeO₃.

As *x* increases in SrMn_{1−*x*}Fe_{*x*}O₃, the energy positions of the L_3 ($2p_{3/2}$) and L_2 ($2p_{1/2}$) peaks do not change and their line shapes are very similar to one another [see Fig. $2(a)$ $2(a)$]. In particular, the line shape of the L_2 peak does not change with *x* in SrMn_{1−*x*}Fe_{*x*}O₃, indicating that the valence state of Fe⁴⁺ ions remains unchanged with *x*. Even though the low-energy shoulder $(h\nu \sim 708 \text{ eV})$ becomes more pronounced in the dilute Fe concentration $(x=0.2)$, its position is different from that of α -Fe₂O₃ ($h\nu$ ~ 707.6 eV) and the line shape of the *L*₂ peak is very different from that of α -Fe₂O₃. Thus, the existence of the trivalent $Fe³⁺$ states is refuted even in the dilute Fe concentration. This conclusion seems to disagree with the suggestion of a possible CD of Fe³⁺-Fe^{[5](#page-4-5)+} in SrMn_{1-*x*}Fe_{*x*}O₃,⁵ which was based on Mössbauer measurements.

As observed for the Mn $2p$ XAS spectra, the Fe L_3 peak becomes narrower as the Fe concentration decreases but keeping the valence states of Fe ions more or less the same. The narrowing of the linewidth might reflect the change in charge transfer from the O 2*p* ligand orbitals to Fe 3*d* orbitals due to the weaker covalency of Fe 3*d* orbitals in the dilute Fe concentration. Then this feature is consistent with the occurrence of the MIT with decreasing *x*.

Figure [3](#page-2-0) compares the partial spectral weight distributions of the Mn and Fe 3*d* states in the valence-band PES spectra $(x=0,1)$ and the O 1*s* XAS spectra of SrMn_{1−*x*}Fe_{*x*}O₃ (0 \leq *x* \leq 1). The Fe and Mn 3*d* PES spectra were determined from the Fe and Mn $2p \rightarrow 3d$ resonant photoemission spectroscopy (RPES) measurements. It is well known that the 3*d* electron emissions of transition-metal (T) ions are enhanced in the T $2p \rightarrow 3d$ RPES.³² We have adopted the extraction procedures for the Fe and Mn 3*d* PES spectra as described in Ref. 33 (see Fig. 5 in Ref. 33 and the related explanation therein).

FIG. 4. (Color online) (a) and (b) The HS electronic configurations for a Fe⁴⁺ ion and Mn⁴⁺ ion, respectively, under the O_h symmetry. (c) and (d) Schematic PDOSs for Fe 3*d* and Mn 3*d* electrons in SrMn_{1−*x*}Fe_{*x*}O₃.

The Fermi level E_F in the valence-band PES spectrum corresponds to 0 eV in binding energy (BE). The Fe 3d PES spectrum of $SrFeO₃$ exhibits a rather broad peak centered at \sim 2 eV below E_F , while the Mn 3*d* PES spectrum of SrMnO₃ exhibits a sharp peak \sim 2 eV below E_F . Since Fe ions are tetravalent $(3d^4)$ in SrFeO₃ (see Fig. [2](#page-1-1)), one can identify the broad peak $(\sim 2$ eV BE) and the weak emission near E_F in SrFeO₃ as the occupied Fe t_{2g}^3 and e_g^1 states, respectively († denotes the majority spin). Similarly, the sharp peak at \sim 2 eV BE in SrMnO₃ is attributed to the occupied Mn t_{2g}^3 \uparrow states for Mn⁴⁺(3*d*³) ions. The broad feature around $6-\bar{7}$ eV in BE for both SrFeO₃ and SrMnO₃ is considered to be the O 2*p* states that are hybridized with the Mn/Fe 3*d* states. The large O 2*p* electron character in Fe and Mn 3*d* PES spectra suggests the strong hybridization between Fe/Mn 3*d* and O 2*p* states in SrMn_{1−*x*}Fe_{*x*}O₃.

The O 1*s* XAS spectrum of a transition-metal oxide is considered to represent the unoccupied T 3*d*, T *sp*, and Sr 4*d* states via the hybridization with the O $2p$ states.³⁴ The lowest-energy peaks in the O 1*s* XAS of SrMn_{1−*x*}Fe_{*x*}O₃ would correspond to the unoccupied $e_g \uparrow$ and $t_{2g} \downarrow$ states of Fe and Mn ions, while the shoulder at the higher-energy side would correspond to the unoccupied *eg*↓. Then the peaks at higher energies are attributed to the unoccupied Sr 4*d* and Fe/Mn 4*s*-4*p* states as labeled in Fig. [3](#page-2-0)(b) (\downarrow denotes the minority spin). This figure reveals that the O 1s XAS spectra change systematically from *x*=0 to 1. Indeed the O 1*s* XAS spectra of the intermediate *x* concentrations can be reproduced by the linear combination of O 1*s* XAS spectra of *x* $=0$ and 1 even though the quantitative ratios of the two XAS spectra are not the same as the nominal concentrations of Fe and Mn ions in SrMn_{1−*x*}Fe_{*x*}O₃. More detailed assignments of these peaks are shown schematically in Fig. [4.](#page-2-1)

Note that the Fe $3d$ PES spectrum for S rFeO₃ reveals the finite spectral weight near E_F while that for SrMnO₃ reveals the negligible spectral weight near E_F . This trend agrees well with the metallic ground state of S rFeO₃ and the insulating ground state of $SrMnO₃$. The large-energy-scale line shapes of both the Mn 3*d* PES and the Fe 3*d* PES are found to be similar for the other *x* concentrations; and the finite spectral weights near E_F are observed for $x > 0.6$ in the Fe 3*d* PES, while the spectral weights near E_F are negligible in the

FIG. 5. (Color online) (a) Comparison of the Mn 2*p* XMCD spectra of SrMn_{1−*x*}Fe_{*x*}O₃. (b) Comparison of the Fe 2*p* XMCD spectra of SrMn_{1−*x*}Fe_{*x*}O₃. All the XMCD spectra were obtained at $T \sim 80$ K.

Mn 3*d* PES. Further, the broader Fe 3*d* PES peak—as compared to the Mn 3*d* PES peak—reflects the larger bandwidth of the Fe 3*d* states than the Mn 3*d* states. In other words, the Fe 3*d* electrons are more itinerant than the Mn 3*d* electrons, which is due to the stronger Fe 3*d*-O 2*p* hybridization than the Mn 3*d*-O 2*p* hybridization. The large Fe 3*d*-O 2*p* covalency is consistent with the absence of the Jahn-Teller distortion in $SrFeO₃$. This conclusion is also consistent with the experimental signature of the increasing covalency of the Fe-O bond for larger *x* in SrMn_{1−*x*}Fe_{*x*}O₃, which was concluded from XRD measurements.⁵

In Figs. $4(a)$ $4(a)$ and $4(b)$, we present the energy-level diagrams for $Fe⁴⁺$ and $Mn⁴⁺$ ions, respectively, based on the findings of Fig. [3.](#page-2-0) Here both Fe and Mn ions are under the O_h symmetry and assumed to have the HS electronic configurations. The corresponding schematic partial densities of states (PDOSs) for Fe 3*d* and Mn 3*d* electrons are shown in Figs. $4(c)$ $4(c)$ and $4(d)$, respectively. Blue and red colors denote the majority-spin and minority-spin states, respectively. The essence of Fig. [4](#page-2-1) is that both Fe and Mn ions are in the tetravalent states so that the topmost occupied states and the lowest unoccupied states are the Fe *eg*↑ states. Consequently, the lowest-energy peak of the O $1s$ XAS of SrFeO₃ is located closer to E_F than that of SrMnO_{[3](#page-2-0)} (see Fig. 3). This is because the unoccupied e_g ^{\uparrow} majority-spin states of Fe⁴⁺ ions are partially filled, whereas the unoccupied *eg*↑ majority-spin states of Mn^{4+} ions are completely empty (well above E_F).

According to the above assignments, the exchange splitting energy J_H is roughly \sim 3 eV for both Fe⁴⁺ and Mn⁴⁺ ions, which is estimated from the energy separations between e_g [↑] (the lowest-energy peak) and e_g ^{\downarrow} (the third peak) in the O 1*s* XAS. In the O 1*s* XAS, the energy separation between the second peak $(t_{2g} \downarrow)$ and the third peak $(e_g \downarrow)$ is roughly \sim 2 eV for both Fe and Mn ions, suggesting that the crystalfield energy 10Dq of Fe and Mn ions is about 10Dq \sim 2 eV.

Figure [5](#page-3-0) compares the Mn and Fe 2*p* XMCD spectra of SrMn_{1−*x*}Fe_{*x*}O₃ (0 ≤ *x* ≤ 1). All the XMCD spectra have been normalized to the area of the corresponding XAS spectra. Our Fe $2p$ XMCD spectrum of SrFeO₃ is similar to that in literature, 31 even though the quality of our XMCD spectrum is worse probably due to the weaker magnetic field employed

in this work $(\sim 0.7 \text{ T})$ than that in Ref. [31](#page-4-28) $(\sim 2 \text{ T})$. The Mn 2*p* states in SrMn_{1−*x*}Fe_{*x*}O₃ exhibit essentially negligible XMCD effect for all *x*.

In contrast, the Fe 2*p* XMCD spectra of SrMn_{1−*x*}Fe_{*x*}O₃ show the enhanced XMCD signals for $0.6 \le x \le 0.8$, even though their absolute intensity is weak. This trend is shown better in Fig. [6,](#page-3-1) which plots the peak-to-peak intensity of the Fe L_3 (2 $p_{3/2}$) XMCD spectra versus *x* in SrMn_{1−*x*}Fe_{*x*}O₃. In this figure, the Fe L_3 XMCD signal for each concentration is normalized to the integrated XAS intensity of the corresponding XAS spectrum, and the error bars represent the finite experimental errors in the estimation of the Fe L_3 XMCD signals. This finding is very interesting because the spin-glass behavior has been observed⁵ in SrMn_{1−*x*}Fe_{*x*}O₃ for $0.6 \le x \le 0.8$. Our XMCD study suggests that the spin-glass behavior in SrMn1−*x*Fe*x*O3 originates from Fe ions but not from Mn ions. Further, the finding of the essentially negligible XMCD effect for the Mn 2*p* states for all *x* supports the tetravalent Mn⁴⁺ states in SrMn_{1−*x*}Fe_{*x*}O₃, in agreement with that of Fig. [1.](#page-1-0)

FIG. 6. (Color online) Plot of the peak-to-peak intensity of the Fe L_3 (2 $p_{3/2}$) XMCD spectra versus *x* in SrMn_{1−*x*}Fe_{*x*}O₃. Error bars are denoted in this figure.

To summarize, the electronic structures of fully oxygenated SrMn_{1-x}Fe_xO₃ (0≤x≤1) have been investigated by using PES, XAS, and XMCD. XAS measurements near the Mn and Fe 2*p* absorption edges reveal that both Mn and Fe ions are nearly tetravalent in SrMn_{1−*x*}Fe_{*x*}O₃ for the whole range of *x*. Thus, no CD of $Fe^{3+} + Fe^{5+}$ is observed in $SrMn_{1-r}Fe_rO₃$ in this study. Valence-band PES reveals that the Mn⁴⁺ states with the t_{2g}^3 occupied configuration are located about \sim 2 eV below E_F . On the other hand, the Fe 3*d* PES spectrum due to Fe⁴⁺ $(t_{2g}^3 \uparrow e_g^1 \uparrow)$ states are broader than the Mn 3*d* PES spectrum. Further, the Fe 3*d* PES spectrum of $SrFeO₃$ exhibits the finite PES intensity near E_F . The latter finding agrees with the metallic ground state of S rFe $O₃$ and both of these findings indicate the stronger Fe 3*d*-O 2*p* hy-

*Corresponding author. kangjs@catholic.ac.kr

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